



Science Arts & Métiers (SAM)

is an open access repository that collects the work of Arts et Métiers Institute of Technology researchers and makes it freely available over the web where possible.

This is an author-deposited version published in: <https://sam.ensam.eu>
Handle ID: <http://hdl.handle.net/10985/17335>

To cite this version :

Christophe FOND, Gilles HOCHSTETTER, Jean-Benoit KOPP - Rapid crack propagation in PA11: An application to pipe structure - Engineering Fracture Mechanics - Vol. 202, p.445-457 - 2018

Any correspondence concerning this service should be sent to the repository

Administrator : scienceouverte@ensam.eu



Rapid crack propagation in PA11: An application to pipe structure

Jean-Benoît Kopp^{a,*}, Christophe Fond^b, Gilles Hochstetter^c

^a I2M, Arts et Métiers Paris Tech, Esplanade des Arts et Métiers, F33800 Talence, France

^b ICube, Université de Strasbourg, 2 rue Boussingault, F67000 Strasbourg, France

^c Arkema France, 420 rue d'estienne d'orves, F92700 Colombes, France

ABSTRACT

Keywords:

Rapid crack propagation
Dynamic fracture
Energy release rate
Polyamide
Pipes
Finite element
Inertial effects
Fracture surface roughness

Dynamic fracture mechanism in Polyamide 11 (PA11) material has been described at laboratory scale to access to an intrinsic material parameter. A liquid transportation application is considered with polymer pipes. A preliminary numerical analysis of the rapid crack propagation (RCP) in polymer pipe is firstly realised. Two boundary conditions, imposed displacement or pressure, are numerically investigated. The work of external forces is not negligible for pressurized polymer pipe. A reliable estimate of the dynamic energy release rate G_{Id} is in this last case not guaranteed. To limit unwanted structural effects a specific experimental device has been used to ensure a permanent regime of RCP in Pre-Stressed Pipe Specimen (PS2). Experimental dynamic fracture tests are realised with Polyamide 11 PS2. Dynamic instabilities inducing “ring-off” and “snake” mechanisms which could appear during full-scale test are not observed with this new test. A finite element procedure is used to estimate the material toughness G_{ID} of PA11. Knowing the crack tip location during RCP inertia effects (i.e. kinetic energy) are quantified. The mean crack tip velocity is observed not to change in PA11 whatever the crack configuration (branching or not). This velocity is known to be the crack branching velocity ($\approx 0.6c_R$). The average dynamic energy release rate $\langle G_{ID} \rangle$ is equal to $1.5 \pm 0.1 \text{ kJ m}^{-2}$ at the crack branching velocity. The non-trivial fracture surface roughness is observed with a scanning electron microscope.

1. Introduction

Studying dynamic fracture in a material is not a simple matter. The main question is probably not why the crack is initiated and how much the energy release rate is necessary for that, but what is the minimal energy release rate necessary to ensure rapid crack propagation (RCP) in the material. If it is possible to answer this last question, it is possible to conceive reliable structures in order to avoid storing enough energy in the structure to ensure dynamic fracture. The minimal energy release rate to ensure RCP is generally significantly inferior to the energy release rate necessary to initiate a crack. That is why dynamic fracture happens even if a safety coefficient is considered. If an external accidental impact provides enough energy to initiate the crack, the dynamic crack propagation is possible in pre-stressed structures. Two international Standard test methods (full- and lab-scale) exist to characterize RCP in polymer pipe but it remains according to Leever's imperfectly understood [1]. We totally agree with this comment.

Polymer materials (polyethylene and polyamide especially) often used to manufacture pipes can suffer to RCP [2–11]. This kind of mechanism which is fortunately infrequent in this kind of structure is highlighted by a longitudinal dynamic crack propagation at some hundred meters per second. The external loading, the crack speed and the material micro-structure are known to affect the crack path. For example, in the case of fluid-pressurized polymer pipelines, standard tests reveal specific mechanisms such as the

* Corresponding author.

E-mail address: jean-benoit.kopp@ensam.eu (J.-B. Kopp).

Nomenclature			
a	the crack length	R	the mean radius of the pipe
\dot{a}	the crack velocity	RCP	rapid crack propagation
B	the sample thickness's	R_e	the external radius of the pipe
c_R	the Rayleigh wave speed	R_i	the internal radius of the pipe
d	refers to strained pipe	S	the beam section
E	the Young modulus	u_0	the dimensionless displacement of the fractured pipe
E_d	the dynamic Young modulus	v_u	the ultra-wave velocity
F	the applied force to deform the pipe radially	W_{dis}	the dissipated energy induced by numerical damping
F_{BT}	the analytical solution of the force F derived from the beam theory	W_{ext}	the work of external forces
G_{IBT}	the energy release rate of a strained pipe derived from the beam theory (structure parameter)	W_{frac}	the energy of fracture
G_{I0}	the quasi-static energy release rate	W_0	the elastic energy before fracture
$G_{I0.01c_R}$	the quasi-static energy release rate obtained numerically for $\dot{a} = 0.01c_R$	δ	the relative displacement of the pipe poles
G_{Id}	the dynamic energy release rate (structure parameter)	ν	the Poisson ratio
G_{ID}	the dynamic energy release rate (material parameter)	ω_0	the free frequency ω_0 of the fractured pipe
G_{IDc}	the critical dynamic energy release rate (material parameter)	ρ	the density of the material
I	the second moment of area of the beam	τ_0	the time associated to the free frequency ω_0 of the fractured pipe
$K_{cin.}$	the Kinetic energy	BT	beam theory
l_n	the notch length	b.c.	boundary conditions
L	the pipe length	p	refers to pressurized pipe
		PA11	polyamide 11
		PS2	pre-stressed pipe specimen
		PMMA	polymethylmethacrylate
		RT-PMMA	rubber toughened polymethylmethacrylate

“ring-off” and a sinusoidal crack path called a “snake” [3,11]. These probable kinds of structural effects could influence, such as viscoplasticity and inertia effects, the dynamic of fracture of polymer materials in a sensitive manner. The estimate of material parameters as the energy release rate or the stress intensity factor is therefore complex [12–14]. These kinds of parameters are necessary to design reliable structures. That is why RCP in materials have been extensively studied for many years especially in mode I and by considering as often as possible the linear elastic fracture mechanics formalism [15–23]. Considering a dynamic regime of propagation in polymer materials the fracture process zone can be often considered confined and therefore negligible comparatively to the crack length. The opening mode (mode I) is commonly studied due to its criticality for materials. In mode I the material needs less energy to maintain crack propagation comparatively to mode II and III.

The use of semi-crystallines and rubber toughened polymers to limit crack initiation and propagation lets appear a specific dynamic fracture behaviour [24,25]. The soft part of the microstructure (i.e. the amorphous part of the semi-crystalline or the elastomeric part in rubber toughened polymers) increases the critical energy release rate to initiate a crack. But this reinforcement is often useless during a high loading rate, for example during a rapid crack propagation. The microstructure reinforcement increases the crack initiation resistance of the material but has no effect on the crack propagation resistance. In the case of rubber toughened polymethylmethacrylate (RT-PMMA), the crack initiation toughness is approximately ten times bigger than that of polymethylmethacrylate (PMMA). But the crack propagation toughness is quite identical between PMMA and RT-PMMA. In RT-PMMA, the bigger the crack length, the lower the energy release rate. That is why the material can be called (as Schirrer proposed [26]) “hyperbrittle”. At high strain rates elastomeric particles are under the well known glass transition temperature (T_g) and only the matrix resistance should be considered [26,27,24]. During a quasi-static loading rubber particles have time to cavitate. This is highlighted by the whitening of the material. During a rapid crack propagation this last mechanism is not observable any more [24]. One could consider RT-PMMA as a model two phased material with a soft and a stiff part. Semicrystallines which have a comparable microstructure could probably highlight the same transition in damage and fracture mechanisms.

It is more or less recently admitted that the dynamic energy release rate G_{ID} of polymer materials varies with the crack tip velocity which is in the range of a fraction of Rayleigh wave speed (c_R) [26,28] or with the created fracture surface area [24,25]. For amorphous polymer material such as PMMA the bigger the energy release rate the bigger the crack tip velocity. For RT-PMMA the crack tip velocity is observed not to change at macroscale during dynamic regime of propagation whatever the energy stored in the structure (i.e. the dynamic energy release rate G_{Id}). This velocity is the well known crack branching velocity at approximately $0.6c_R$ [15]. Crack branching due to inertia effects is therefore current for these kinds of rubber-toughened materials as it is in semi-crystalline. The modification of the microstructure with the introduction of micro-metric rubber particles is efficient to absorb impact energy at intermediate impact velocity (approximately less than 10 m s^{-1}). But it induces complex material behaviour. Micro-metric particles are known to modify stress field in the crack front. This could explain *frustrated microscopic branching events which are at the origin of non-trivial fracture surface* according to Sharon and Fineberg [29]. A “loss of uniqueness” of G_{ID} at the crack branching velocity is revealed. This is due to the non consideration of the amount of created fracture surface in the estimate of G_{ID} . For these

kinds of rubber toughened materials as in semi-crystallines the bigger the energy release rate the bigger the amount of created fracture surface.

In this paper, a first part will describe the preliminary numerical structural analysis of RCP in polymer pipe. A second part is devoted to describe the experimental procedure to ensure RCP in PA11 pipes. The material behaviour during RCP is described as the crack tip location as a function of time which is recorded. The evolution of G_{ID} as a function of the crack tip velocity and the fracture surface roughness is finally discussed.

2. A preliminary numerical structural analysis

The finite element method is used to estimate the dynamic energy release rate during RCP in polymer pipes. The numerical procedure consists (1) in predicting the structure behaviour and (2) analysing experimental data with access to material parameter, i.e. the dynamic fracture energy. Boundary conditions, the structure geometry and the crack velocity significantly influence the dynamic behaviour of the structure and also the estimate of G_{Id} . In this part two boundary conditions are considered for modelling the moving crack in a pipe: (a) a pre-stressed pipe with imposed displacement at the poles and (b) a uniformly liquid pressurized pipe. The available energy stored in the structure for crack propagation (i.e. G_{Id}) is estimated as a function of boundary conditions, crack velocity and pipe geometry. In the part 3.1, the numerical procedure is only used to analyse experimental data. The experience allows to access the crack path history (crack length as a function of time $a(t)$) during RCP and the elastic energy stored in the structure before crack initiation and propagation. The numerical model allows to estimate the energy released by the structure to the material (G_{ID}) to ensure RCP taking into account inertia effects.

2.1. Finite element mesh

A model of an half pipe is meshed. A linear elastic behaviour is considered for the material. The symmetry of the structure is used for fracturing. 8-nodes and 20-nodes cubes have been used. For a good compromise between accuracy and computation time, each element corresponds to an angle of 2 degrees. 3 layers of 8-nodes elements in the thickness of the shell give relatively accurate results equivalent to a single layer for 20-nodes elements (see here [30] to have more details). The longitudinal length, defining the thickness of an elementary slice of the pipe, corresponds to 1.67–3.33 times the thickness of the shell. The total length of the pipe for the numerical results presented herein is 20 times the mean radius R , $R = (R_i + R_e)/2$, of the pipe. As shown in previous analysis concerning mainly the steady state regime of a dynamic fracture [14,26], the use of special fracture elements at crack tips is not necessary to compute the energy release rate. In fact, the energy release rate is computed by differentiating the elastic energy integrated on the whole structure. As the geometry ensures a quasi-steady state regime of propagation, it is assumed that a specific treatment of the singularity is not necessary since the error done concerning the energy integration at crack-tip singularity is eliminated by the differentiation. The element type is selected by a modelling of the dynamometric ring problem. Firstly, the analytical solution is computed. Considering quasi-static loadings derived from the beam theory (BT), Eq. (A.1) gives the solution of the force as a function of displacement for the dynamometric ring. This solution assumes a linear elasticity. This analytical solution (F_{BT}) has been previously compared to the numerical solution (F) (see in [30]). One has to keep in mind that according to the S^T Venant hypothesis, pressures applied inside the pipe, in the case of imposed displacement at the poles, can be replaced by concentrated forces, and according to the Navier-Bernoulli assumption, plane sections remain plane.

2.2. Boundary conditions

Part of boundary conditions are taken to avoid rigid body displacements. The initial imposed null displacements along the crack path are then withdrawn to simulate the crack propagation. Two loading types are considered before the fracture in pipe structure (see in Fig. 1). The first one considers imposed parallel displacements at the poles of the pipe with variable widths. The second one considers a uniform pressure inside the pipe. The latter one is a more or less realistic point of view only for crack running at speed larger than the decompression waves speed in the liquid ensuring the pressure. For smaller crack speed a complex interaction

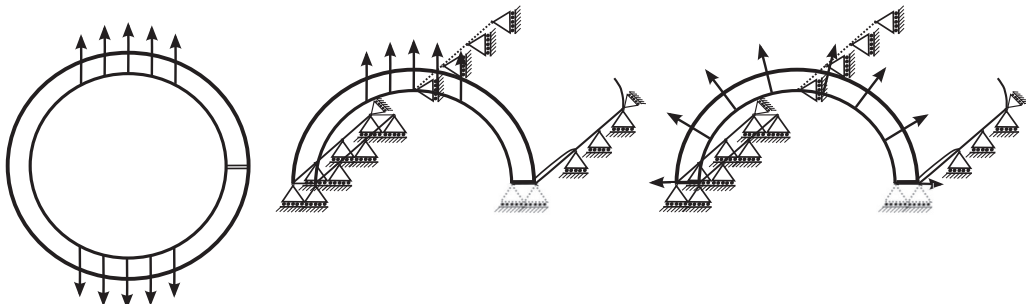


Fig. 1. Boundary conditions ensuring symmetry for imposed displacements and internal pressure.

between liquid and solid should be accounted for, including the loss of liquid through the opened crack. This last point induces heavy problems to solve and is not accounted herein. Nevertheless, it is noticeable that assuming a non-variable pressure during fracture gives an upper bound to the work done by external forces, i. e. by pressure in the deformation of the pipe.

2.3. Validation of the moving crack model

As presented here [30] the numerical procedure has been validated using the analytical solution of Broberg [16]. Fig. 2 shows numerical results obtained for a crack speed corresponding to about the upper limit of the quasi-static cases, i. e. 10% c_R . Imposed displacements are considered as boundary conditions. It is shown that the energy release rate predicted from quasi-static BT for infinitely large pipe (cf. Appendix A) is well retrieved at the plateau value in the range of $[0.2L, 0.9L]$. One can also notice as expected that the dissipated energy $W_{dis.}$ associated to the damping ensuring the stability of numerical computations is negligible compared to the energy of fracture $W_{fract.}$ or equivalent to the elastic energy before fracture W_0 .

2.4. Modelling of the moving crack in a pipe pre-stressed by imposed displacements at the poles

A typical finite element model of the fracture propagation is shown during RCP here [30]. The numerical simulations were applied for pipes of 3 different wall thicknesses with different crack tip velocities. It appears that a quasi constant energy release rate can be obtained after a length larger than 5 times the external radius R_e . Fig. 3 shows the numerical results of the evolution of the dynamic energy release rate as a function of the normalised crack length for three R/B ratios (4.83, 9.67 and 19.3) and four crack tip velocities (0.01–0.1–0.2–0.3 c_R). Due to the border effect, the energy release rates increase when the crack tip approaches the opposite border within a distance of about 2 times of R_e . When the crack tip velocities increase, the energy release rate as expected, decreases. The influence of the wall thickness of pipes can also be observed. The thinner the wall, the lower the energy release rate. This effect is associated to the opening speed of the fracture structure (fracture ring), which is linked to the kinetic energy $K_{cin.}$. The bigger the opening speed, the bigger the kinetic energy, and the less energy released.

2.5. Modelling of the moving crack in a pipe under pressure

The modelling of the moving crack in a liquid-pressurized polymer pipe is investigated. It is more complex as boundary conditions point of view than the pre-stressed pipe but more representative of the real application and standard tests. As a material point of view, the critical case is considered. Indeed the internal pressure is applied to be constant even during the dynamic propagation, i.e., the main crack is running faster than the decompression waves in the liquid and no leak of the liquid is modelled during dynamic fracture. According to the numerical results, the dynamic energy release rates are compared to the quasi-static energy release rate G_{I0} which is computed without the variation of the work of external forces (cf. Appendix B). As the results of the pre-stressed pipe, the

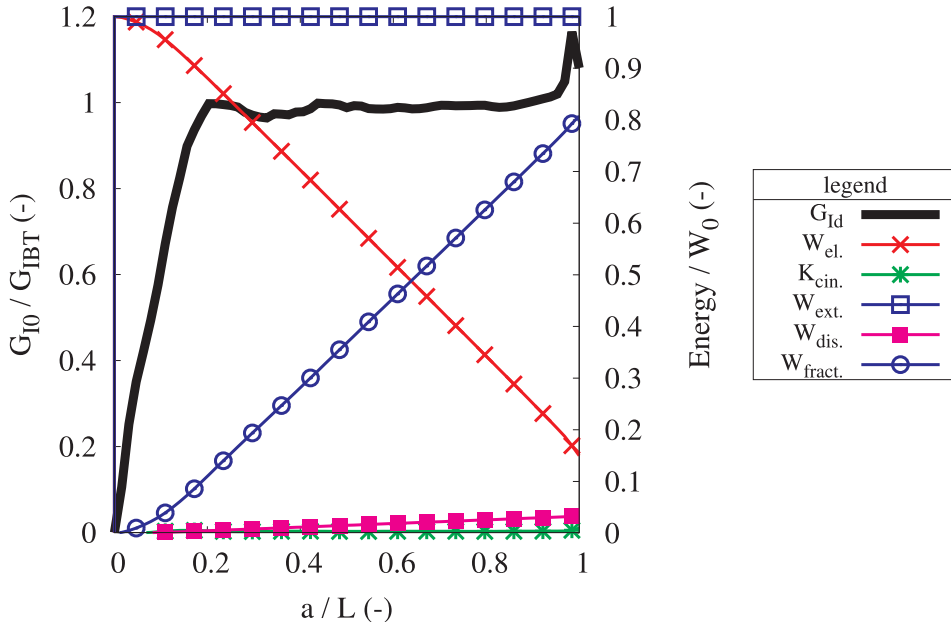


Fig. 2. Energy release rate and energies vs. crack length for a pipe of dimensions $R = 4.83B$ and $L = 20R$ (imposed displacement limit conditions). The total length L of the pipe for the numerical results presented herein is 20 times the mean radius R , $R = (R_i + R_e)/2$, of the pipe with R_i and R_e respectively the internal and external radius of the pipe. B is the thickness of the pipe. G_{IBT} is the analytical solution of the energy release rate obtained with the help of the BT.

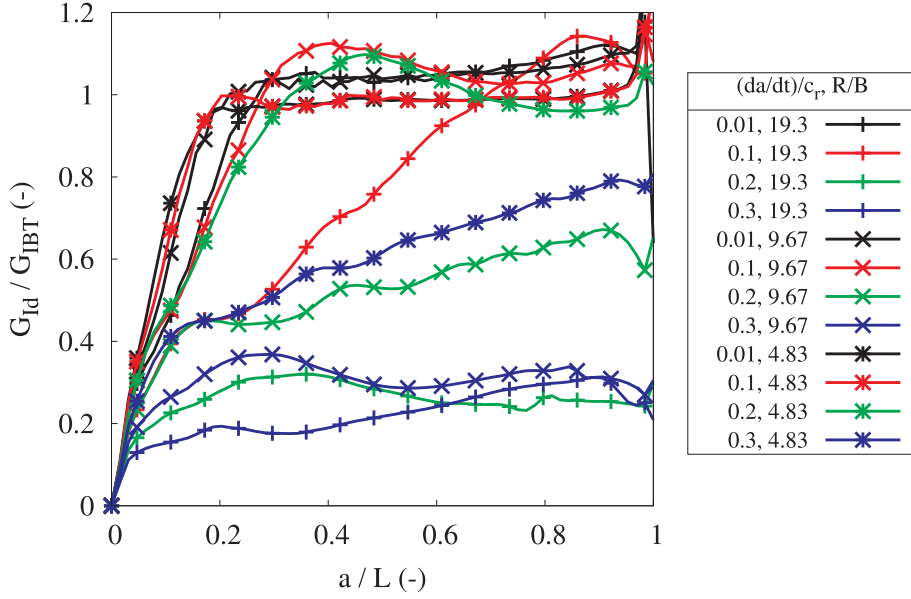


Fig. 3. Dynamic energy release rate values vs. normalised crack length for various crack tip velocities and wall thickness B in a strained pipe (imposed displacement boundary conditions).

energy release rates stabilize after a crack length of $5R_e$ and decrease with the crack velocity (see in Fig. 4). As explained here [30] the dynamic energy release rate G_{Id} of the quasi-static ($0.01c_R$) crack is about 300 times of G_{I0} . This means that the elastic energy released from that stored in the structure becomes a second order parameter in the energy balance sheet. The value of G_{Id} is in the case of pressurized pipe dominated by the work of external forces. The influence of the wall thickness is also studied. Fig. 4 shows the evolution of G_{Id} for 3 different pipe wall thicknesses. A similar tendency for the strained pipes (pre-stressed by imposed displacements) can be observed, the energy release rate increases with the wall thickness.

2.6. Free frequencies of the fractured ring

As shown in the simulation results, it appears clearly that the opening speed of the fractured pipe influences considerably the dynamic energy release rate G_{Id} . This can be explained by linking the free frequencies of the fractured pipe to the dynamic correction

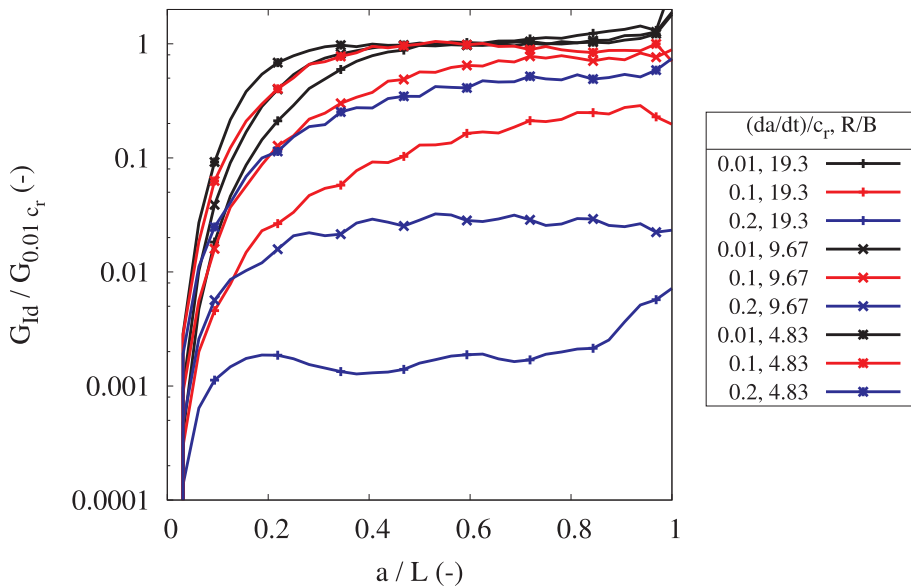


Fig. 4. Dynamic energy release rate values vs. crack length for various crack tip speeds and thickness B in a pressured pipe (imposed load boundary conditions). $G_{0.01c_R}$ is the energy release rate of the quasi-static crack ($0.01c_R$).

factors. A fractured ring is therefore modelled to study the free frequencies of the fractured pipes (see Fig. 5). For a straight beam, the well known dynamic equation of equilibrium takes the form $\rho \ddot{y}S + EI \frac{\delta^4 y}{\delta x^4} = 0$. For a fractured ring which can be considered as a curved beam, the dynamic equation of equilibrium is not so simple but the free frequencies are still linked to $\omega_0 = \sqrt{\frac{EI}{\rho S}}$. Considering a rectangular section beam, ω_0 becomes $\frac{B}{2\sqrt{3}} \sqrt{\frac{E}{\rho}}$. To analyse dimensionless values, one can use the time associated to the free frequency of such a beam which is $\tau_0 = \frac{R^2}{\omega_0}$. For dimensionless displacements, $u_0 = \frac{FR^3}{EI}$ is used, F being the force transmitted before fracture corresponding to $\frac{F_{BT}}{2}$ (cf. Eq. (A.1)) for imposed displacements at poles, or $\langle \sigma \rangle BL$ for imposed internal pressures (cf. Appendix B).

Fig. 6 shows the free-frequencies of a pre-stressed or pressurized pipe. However, for the pressurized one, whatever the wall thickness, the free-frequencies of a fractured ring are more or less the same. For a pre-stressed one, the thinner the wall thickness is, the faster the opening velocity of the fractured ring is, and the more important the dynamic correction is. These observations confirm the deduction from numerical results.

2.7. Discussion

The “dynamic correction factors” presented in Fig. 7 correspond to the total fracture work in the dynamic regime divided by the total fracture work for a crack velocity equal to $0.01c_R$, i. e. to the quasi-static total fracture work. Fig. 7 shows results for imposed displacements and pressure for three wall thicknesses B . It appears clearly that the “structural” energy release rate G_{ld} is strongly dependant on the wall thickness. The thinner the wall is, the bigger the opening of the fractured ring is and the lower the energy release rate is. This effect is boosted by the boundary conditions in pressure case. The energy released is directly linked to the opening speed of the fractured ring. For the sake of simplicity, our model ensures a constant pressure in the pipe which is not realistic for crack velocities lower than the compressive waves speed in the liquid inside the pipe. Nevertheless, even with this simplification, the problem of analysing dynamic fracture in this case is shown to be complex and strongly dependant on the crack velocity and on the

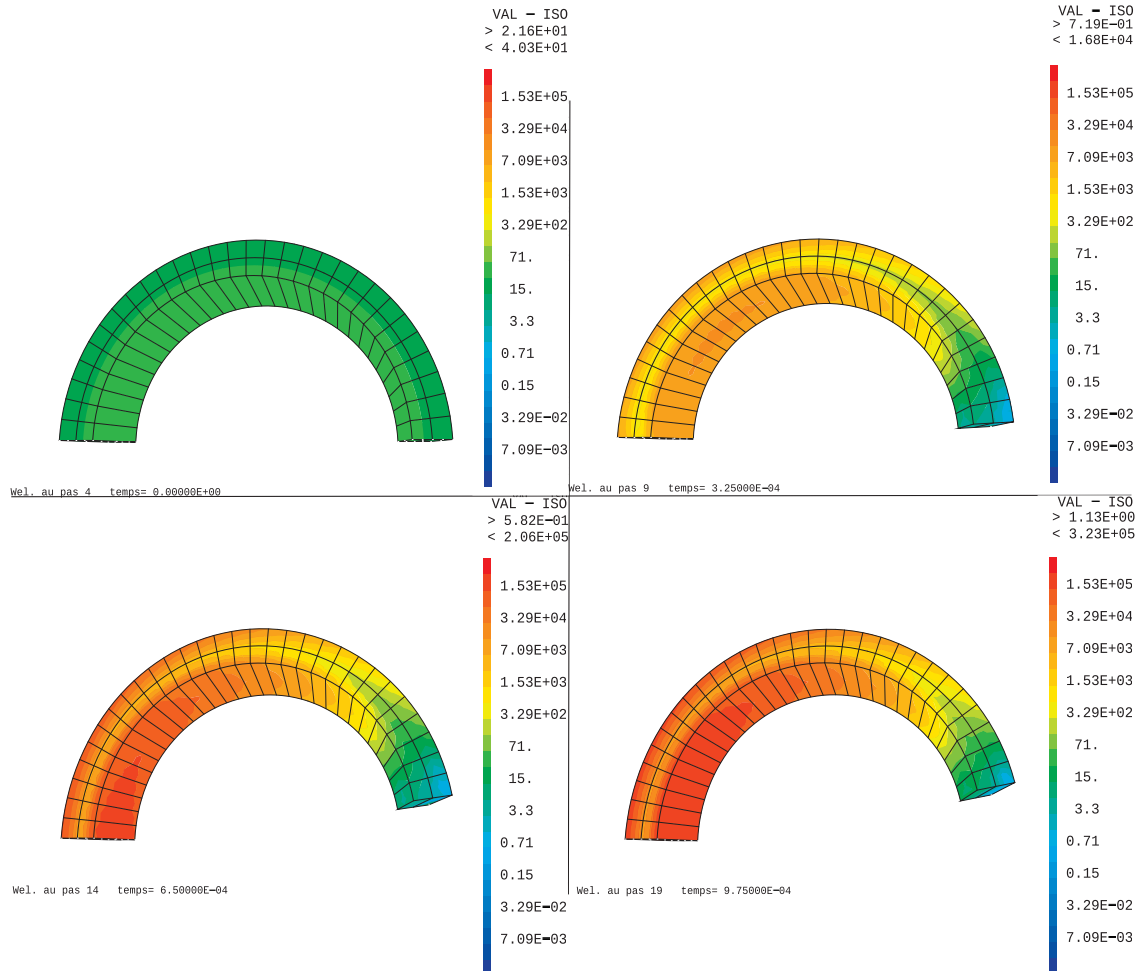


Fig. 5. Isovalues of the elastic energy density during opening of a instantaneously fractured ring.

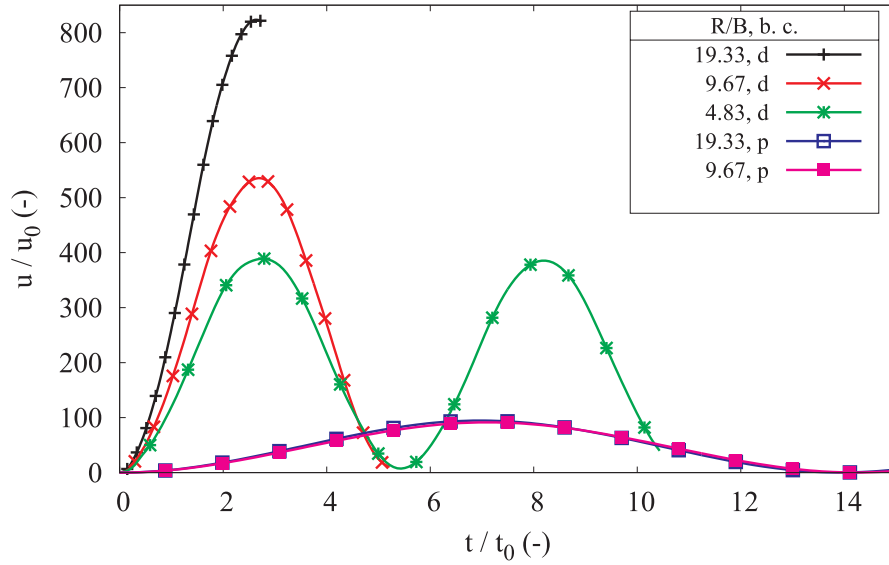


Fig. 6. Normalised displacements vs. normalised time for opening a fractured ring. In the legend, *d* refers to imposed displacement et *p* to imposed load.

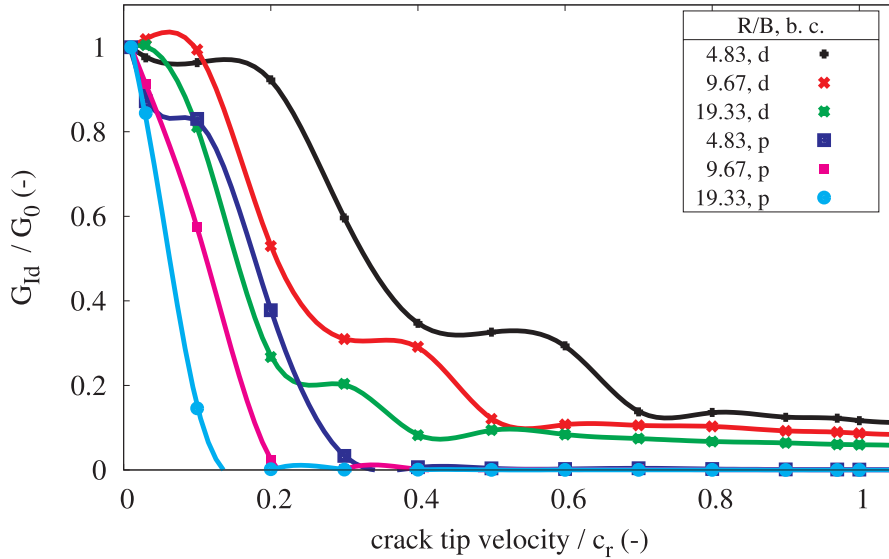


Fig. 7. “Dynamic correction factors” vs. crack velocities for various wall thickness for boundary conditions (b. c.) corresponding to strained (d) and pressured pipes (p).

wall thickness. It is expected that the energy released is also strongly dependant on the pressure decrease and distribution inside the pipe. Hence, it can be concluded that tests with internal gas pressure are not well suited for characterizing the dynamic fracture of the material [5,6].

3. Experimental investigations for cracks running in strained pipe

3.1. Experimental set-up

Based on these numerical results a specific experimental set-up has been used to ensure RCPs in PS2 (see Fig. 9) [30]. The loading method of PS2 is presented in Figs. 8 and 9. The main objective is to pre-stress as uniformly as possible polymer pipes with imposed interior relative displacements (δ) to increase the stored elastic energy in the pipe. The loading system is a combination of three serial modules separated by spacers presented in Fig. 9-up left. The loading system is first adjusted to R_i . It is then introduced in the pipe and is constrained by friction between the internal wall of the pipe and metallic pieces. The global rotation of the loading system is

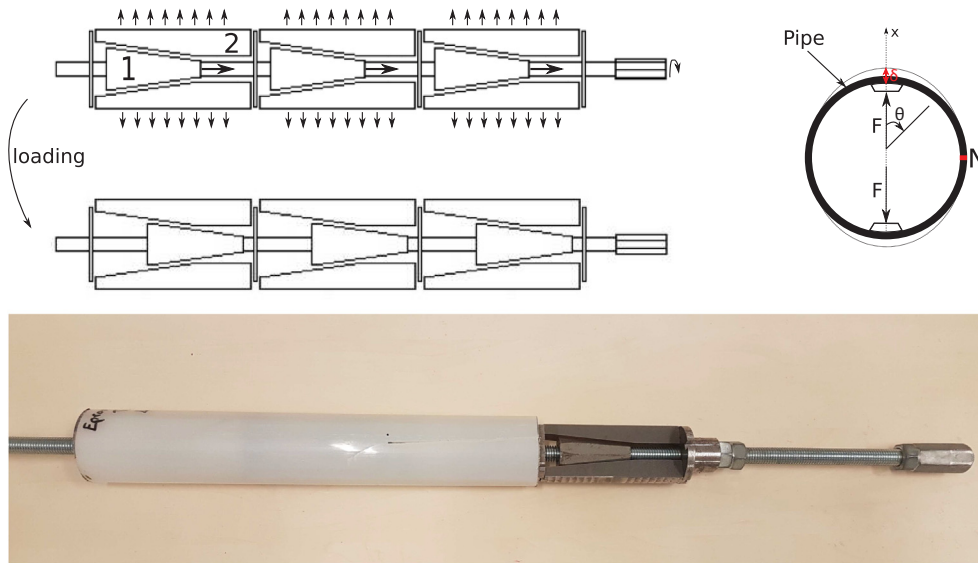


Fig. 8. Principle of pre-stressed pipe test developed at the laboratory. The load is distributed on approximately 18% of the circumference of the pipe with the help of a pair of rigid part-cylinders (2) displacements δ which are ensured by the cone (1) translation on the threaded rod. The notch N is positioned at $\theta = -\frac{\pi}{2}$.

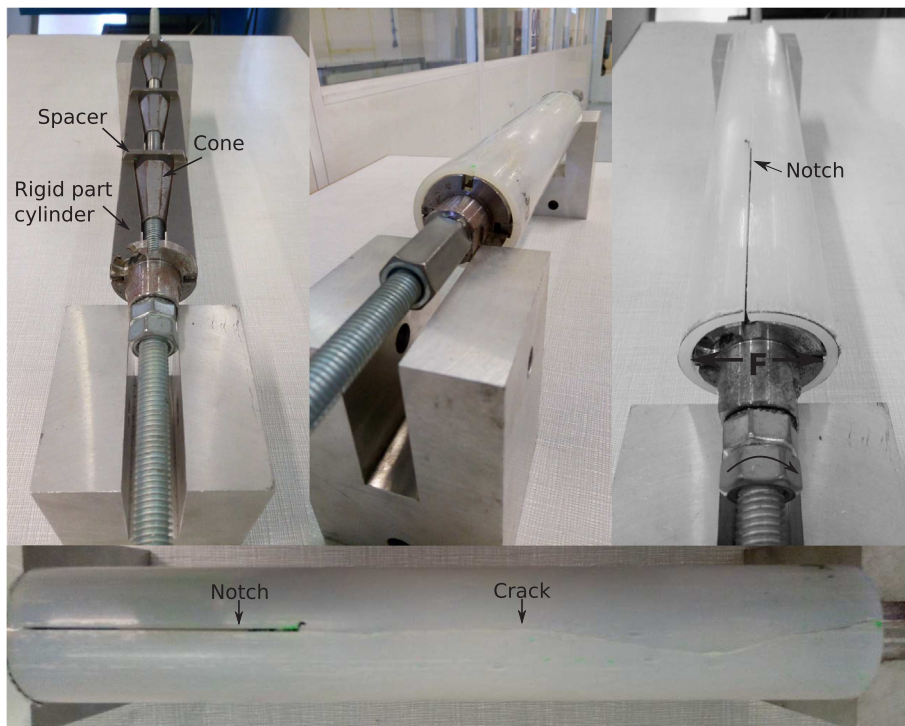


Fig. 9. Captures of the experimental set-up with and without pipe around the loading system.

therefore not possible. The loading system with the pipe is then clamped on a base. Radial force F is applied to the internal surface of the pipe wall by a pair of rigid part-cylinders (2), each of which loads approximately 18% of the circumference. The two part-cylinders are driven outwards by steel cones (1) which, while prevented from rotating by friction, are driven axially along a threaded rod rotated manually by the experimenter (see Fig. 8). The pre-stress of the pipe is stopped and maintained at constant δ by stopping the rotation of the threaded rod. A mode I fracture test is considered. It is noted that this loading system at the poles induces bending at the opposites poles (90° and 270° points). The circular pipe begins to deform therefore in an ellipse during the loading. A significant relaxation time (approximately 15 times the loading time) is considered before initiation. The crack is then initiated artificially with

the help of an external impact on a razor blade in contact with the notch tip (see in Figs. 10-right). The notch is positioned at $\theta = -\frac{\pi}{2}$ compared to the poles. The quasi-static energy release rate G_{I0} can be analytically estimated with the help of Eq. (A.2) or numerically with a global energy balance. The crack path history is captured with the help of a high speed camera (Photron 106 APX-SA5) at 40000 frames per second with a resolution of 704×256 . Knowing the average crack tip velocity $\langle \dot{a} \rangle$ a numerical estimate of the consumed energy by the material G_{ID} to ensure a RCP is possible by taking into account inertia effects.

3.2. Material

The material under study is a polyamide 11 BESNO TL grade provided by Arkema. It was supplied as initially extruded pieces of pipe. The PA11 under study is a semi-crystalline polymer. A degree of crystallinity of 22% has been measured by Differential Scanning Calorimetry analysis for PA11 pipe. The dynamic elastic modulus E_d (see Eq. (1)) has been obtained by ultra-wave analysis and is equal to 1620 ± 82 MPa with $\nu = 0.43$ and $\rho = 1040 \text{ kg m}^{-3}$. The mean ultra-wave velocity $\langle v_u \rangle$ measured in a pipe is equal to $2100 \pm 46 \text{ m s}^{-1}$. The standard deviation has been considered for 10 measurement made on 5 different pipes. The viscoelastic behavior of the polymer is approximated with the only dynamic elastic modulus E_d as described here [31].

$$E_d = \rho v_u^2 \frac{(1 + \nu)(1 - 2\nu)}{1 - \nu} \quad (1)$$

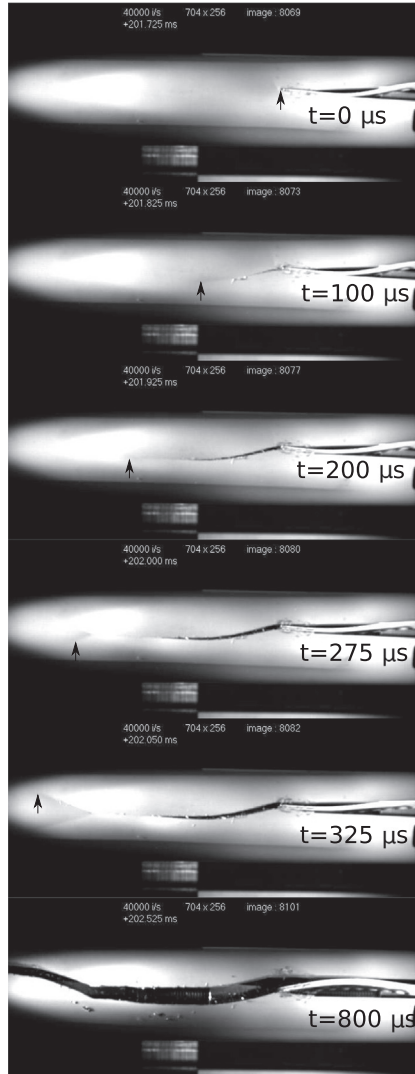


Fig. 10. Typical images of high speed camera with a $4 \cdot 10^4$ images per second sampling rate during rapid crack propagation in polymer pipe ensured with the help of an experimental set-up. Time $t = 0 \mu\text{s}$ is chosen like reference point where the crack begins to propagate in dynamic state. At time $t = 325 \mu\text{s}$ the pipe is almost completely fractured. The arrow highlights the crack tip location during RCP. At $t = 275 \mu\text{s}$ a macroscopic crack branching is observable. At $t = 800 \mu\text{s}$ fragments are caused by the sprayed paint which is used to highlight the image contrast.

3.3. Experimental results

Fig. 10 presents a dynamic fracture of polyamide pipe (B4-see in Table 1), recorded using a high speed camera. The entire test is performed at room temperature. Five pipes are fractured with the help of the experimental device. Pipe dimensions are: $L = 300$ mm, $b = 4$ mm, $R_i = 21$ mm and the notch length $l_n \approx 5R_c \approx 120$ mm. The notch length is defined with the help of numerical results. A quasi-permanent regime of propagation is supposed to be reached. The mean measured crack tip velocity at macroscopic scale $\dot{a} \approx 420 \pm 20 \text{ m s}^{-1} \approx 0.6c_R$ before and after branching. It is deduced at this scale that the crack velocity does not change. The crack propagates therefore at a relatively constant macroscopic velocity. A crack branching event is seen before $t = 325 \mu\text{s}$ in Fig. 10. The crack path is known to depend on the available energy. The bigger the available energy, the bigger the dissipated energy by creating fracture surface. In Fig. 10 the available elastic energy stored in the structure is enough to propagate one straight crack with two branches. Inertia effects are supposed to induce branching at $0.6c_R$ as demonstrated by Yoffe [15]. $\langle G_{I0} \rangle$ is estimated with the help of Eq. (A.2) to be equal to approximately $7.5 \pm 0.9 \text{ kJ m}^{-2}$. The amount of fracture surface area has been calculated as the thickness times the crack length for each sample. At $0.6c_R$ with an averaged dynamic correction of 0.2 estimated with the help of numerical results (see Fig. 11), the mean dynamic fracture energy $\langle G_{ID} \rangle = 1.5 \pm 0.1 \text{ kJ m}^{-2}$.

3.4. Fracture surface analysis

Fracture surface analyses have been realised with the help of a scanning electron microscope. B1 and B3 samples are presented in Fig. 12 at two different scales. At macroscopic scale, it is observed that the fracture surface roughness is different between each samples. B1 concerns a straight crack propagation contrary to B3 which has been observed just before a crack branching. The fracture surface seems to be rougher at the bottom of the image in B3 than in B1. The evolution of the roughness in the thickness is not homogeneous. This highlights as expected that the stress state in the thickness is not uniform. This is due to the loading system which induces radial force only at two opposite poles. The two other poles are free to move inwards. It is difficult to see in these images if the crack front on the inner surface propagates faster than the crack front on the external surface. Crack arrest marks should be analysed to observe it. At microscopic scale (down) the spherulitic microstructure of the PA11 is highlighted. At this scale the crack path is non-trivial. The crack propagates through the spherulites which can probably modify the stress field at the crack tip. These analyses show that the LEFM approximation of the fracture surface as a mean plane is not relevant. The fracture surface roughness and therefore the fracture surface area should be considered in the estimate of G_{ID} . The critical dynamic energy release rate G_{IDc} could then be estimated. By considering the amount of projected surface area (B times a) G_{ID} is overestimated. A relevant estimate of the minimal value of G_{ID} is not possible.

4. Discussion and conclusions

It has been shown that the analysis of the dynamic fracture mechanism in polymer pipe is a rather complex matter and depends strongly on the boundary conditions, the relative wall thickness of the pipe and the crack tip velocity. A preliminary numerical structural analysis has been made to study these effects which are insensitive of elastic material parameters. As explained here [30], the dynamic energy release rate G_{Id} is significantly different between a crack running in pre-stressed or pressurized pipe. In pressurized pipe, the work of external forces dominates in the Griffith energy balance. That is why, for $\dot{a} \approx 0.01c_R$ the dynamic energy release rate G_{Id} is about 300 times of G_{I0} . A relevant estimate of G_{Id} for a pressurized pipe is not guaranteed. The estimate of G_{Id} is moreover strongly dependant on the relative wall thickness whatever the boundary conditions are:

- For the pressurized pipe, whatever the wall thickness, the free-frequencies of a fractured ring are more or less the same. For a pre-stressed one, the thinner the wall thickness is, the faster the opening velocity of the fractured ring is, and the more important the dynamic correction is.

Based on these results a specific experimental set-up is used for studying RCP in PA11 PS2. Imposed displacement boundary conditions allow to have an approximately $L \approx 13R$ permanent dynamic regime of propagation i.e. G_{Id} is quasi-constant during dynamic crack propagation. Some conclusions can be made on the material behaviour:

- The crack tip velocity is observed not to change ($\dot{a} \approx 0.6c_R$) at macroscopic scale whatever the available stored energy in the

Table 1
Quasi-static and dynamic energy release rates for PA11 pipes.

Sample	Temperature (°C)	δ (mm)	$\langle \dot{a}/c_R \rangle$	G_{I0} (kJ/m ²)	G_{ID} (kJ/m ²)
B1	17.5	6.31	0.56	7.0	1.4
B2	17.4	6.69	0.54	7.9	1.6
B3	18.0	6.51	0.62	7.5	1.5
B4	17.9	6.60	0.63	7.7	1.5
B5	18.1	6.42	0.61	7.3	1.5

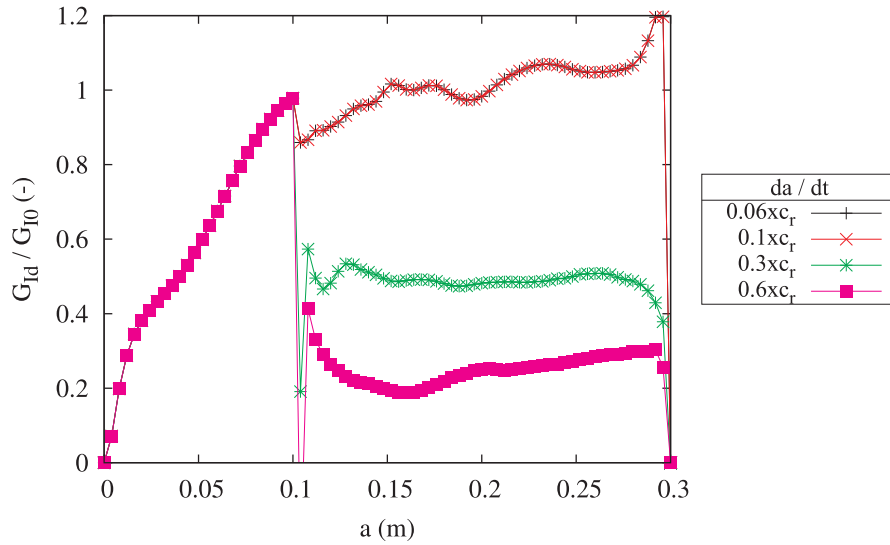


Fig. 11. Evolution of the energy release rate G_{Id} as a function of the crack tip velocity and the crack length for a strained pipe.

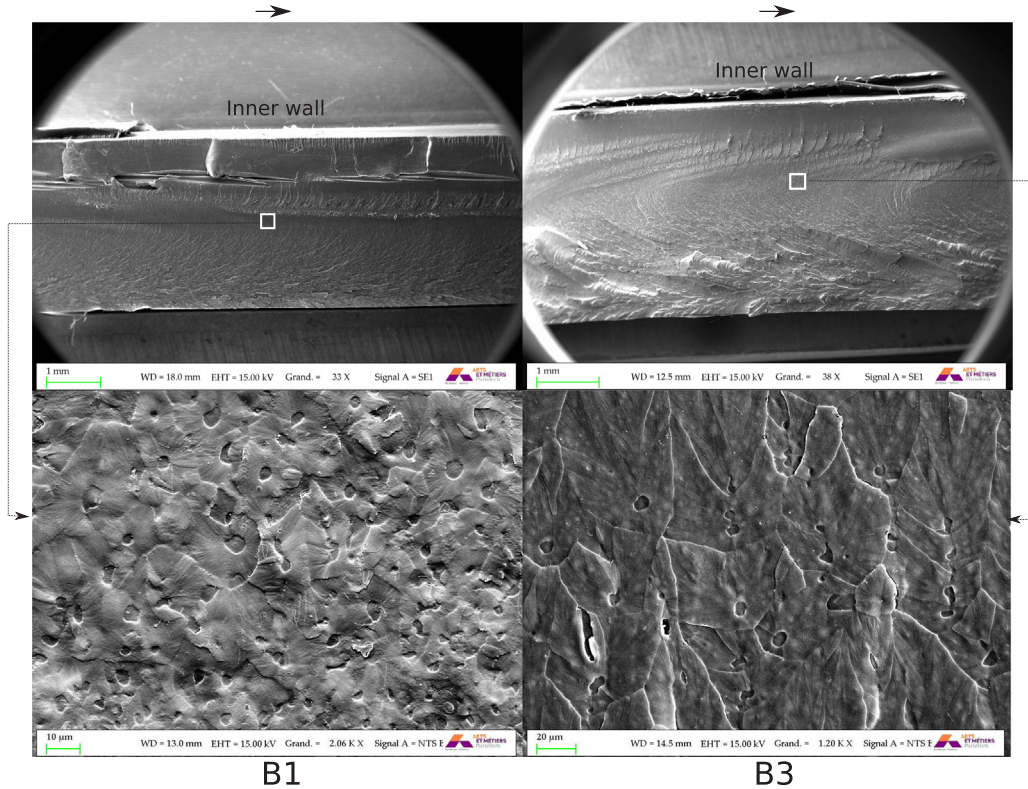


Fig. 12. Typical images of fracture surface roughness of PA11 obtained with a scanning electron microscope. Samples B1 and B3 have been fractured with the pipe loading device presented herein in Fig. 8. The arrow defines the crack propagation direction.

structure before initiation and whatever the crack configuration (straight crack or crack branching).

- Fracture surface analyses suggest that the crack path is not trivial and different between each sample. As for RT-PMMA, it seems that the lower the energy release rate the smoother the fracture surface. The microscopic crack velocity should therefore probably change as a function of the available energy release rate. The bigger the energy release rate the bigger the microscopic crack length the bigger the microscopic crack velocity.
- A relevant estimate of the critical dynamic energy release rate G_{IDc} is difficult since the amount of created surface area is not considered. A more appropriate estimate of G_{ID} as a function of the amount of created fracture surface (and not the thickness times

the crack length) should be considered as it has been suggested for rubber toughened polymer [24,25] to finally access the critical dynamic energy release rate G_{IDc} .

This kind of approach to limit unwanted structural effects has lead us to pre-stress pipe with imposed displacement contrary to full- and lab-scale tests which consider a liquid-pressurized pipe. This controlled solicitation allows to reveal the relevant material behaviour. In the case of RCPs in PS2 PA11 it is observed that contrary to full- and lab-scale tests:

- the crack tip velocity is bigger than the decompression wave velocity, i.e. structural effects do not affect the dynamic of crack propagation with a limited velocity which will not be a ratio of the Rayleigh wave speed. That is why in this case the crack branching velocity is reached.
- “Ring-off” and “snake” mechanisms do not appear during dynamic propagation.

Appendix A. Dynamometric ring model

Beam theory allows to obtain a quasi static reference analytical solution corresponding to the steady state regime in a pipe of infinite length, i. e. without any border effect. Considering a pipe of internal radius R_i and external radius R_e the force F to apply to deform the ring with a δ relative displacement of the poles is given by:

$$F_{BT} = \frac{4\pi EI\delta}{(\pi^2-8)R^3} \quad (A.1)$$

where E is the Young's modulus of the material, I the second moment of area of the beam and R the mean radius of the pipe. I is equal to $(R_e-R_i)^3 b/12$ where b is the thickness of the ring and R is equal to $(R_i + R_e)/2$. The elastic energy is thus equal to $1/2 F_{BT}\delta$ for the entire ring. The energy release rate in the steady state regime is obtained by considering that an increase of Δa of the crack length in an infinitely long pipe corresponds to the passage of a slice of length Δa far in front of the crack tip far below the crack tip. The energy release rate G_I is thus equal to released energy divided by the created surface, i. e. $(1/2 F_{BT}\delta)/(\Delta a (R_e-R_i))$ where I is equal to $(R_e-R_i)^3 \delta a/12$.

$$G_{IBT} = \frac{8\pi E (R_e-R_i)^2 \delta^2}{3(\pi^2-8)(R_e + R_i)^3} \quad (A.2)$$

Appendix B. Pipe under pressure

Considering a pipe under an internal pressure p , the stress state varies only with r in axial coordinates r, θ, z . For thin shells, typically for $(R_e-R_i) \ll R_i$, the variation with r is not significant and it can be assumed at first glance that the stress fields are uniform. An approximate solution for the stress state is then easy to derive from equilibrium. The average stresses, an average value being denoted $\langle \sigma \rangle$, are $\langle \sigma_{\theta\theta} \rangle = \frac{pR_i}{(R_e-R_i)}$, $\langle \sigma_{rr} \rangle \approx \frac{-p}{2}$ and $\langle \sigma_{zz} \rangle = \frac{-pR_i^2}{R_e^2-R_i^2}$. Assuming linear isotropic elasticity, the orthoradial strain is equal to $\langle \epsilon_{\theta\theta} \rangle = (\langle \sigma_{\theta\theta} \rangle - \nu \langle \sigma_{rr} \rangle - \nu \langle \sigma_{zz} \rangle)/E$. The elastic energy for a pipe of length b is equal to $W_{ela.} = \pi R_i^2 p \langle \epsilon_{\theta\theta} \rangle b$ before fracture. The part of the total released energy associated to these elastic energy is equal to $G_{I0} = \frac{W_{ela.}}{(R_i-R_e)b}$. G_{I0} is not the energy release rate since during fracture both the elastic energy, the work of external forces and the kinetic energy will vary. These variations depend on the crack tip velocity – or more exactly location vs. time – and on the loss of pressure associated to the opening of the crack. The last point presents considerable difficulties for numerical modelling. Nevertheless G_{I0} is useful to compare the order of magnitude of work done by external forces during a test to that of the energy available for fracture at initiation of the crack. For the numerical simulations presented herein, the stress $\langle \sigma_{\theta\theta} \rangle = 0$.

$$G_{I0} = G_{Ip} = \frac{\pi p^2}{E} \left(\frac{R_i^3}{(R_e-R_i)^2} + \frac{\nu R_i^2}{2(R_e-R_i)} + \frac{\nu R_i^4}{2(R_e^2-R_i^2)(R_e-R_i)} \right) \quad (B.1)$$

In the case of negligible radial and longitudinal stresses, i. e. $\sigma_{\theta\theta} \gg \sigma_{rr}$ and $\sigma_{\theta\theta} \gg \sigma_{zz}$, an approximate solution gives:

$$G_{I0} = G_{Ip} \approx \frac{\pi p^2 R_i^3}{E (R_e-R_i)^2} \quad (B.2)$$

References

- [1] Leevers P. An engineering model for rapid crack propagation along fluid pressurized plastic pipe. *Eng Fract Mech* 2012;96:539–57.
- [2] Mason J, Chen J. Establishing the correlation between s4 and full scale rapid crack propagation testing for polyamide-11 (pa-11) pipe. *Plastic pipes XIII* 2006.
- [3] Shannon RWE, Wells AA. Brittle crack propagation in gas filled pipelines—a model study using thin walled unplasticised pvc pipe. *Int J Fract* 1974;10(4):471–86.
- [4] Kanninen M, O'Donoghue P. Research challenges arising from current and potential applications of dynamic fracture mechanics to the integrity of engineering structures. *Int J Solids Struct* 1995;32(17):2423–45.
- [5] Greig J, Leevers P, Yayla P. Rapid crack propagation in pressurised plastic pipe-I. Full-scale and small-scale rcp testing. *Eng Fract Mech* 1992;42(4):663–73.
- [6] Yayla P, Leevers P. Rapid crack propagation in pressurised plastic pipe-ii. critical pressures for polyethylene pipe. *Eng Fract Mech* 1992;42(4):675–82.

- [7] O'Donoghue P, Kanninen M, Leung C, Demofonti G, Venzi S. The development and validation of a dynamic fracture propagation model for gas transmission pipelines. *Int J Press Vessels Pip* 1997;70(1):11–25.
- [8] Zhuang Z, O'Donoghue P. Determination of material fracture toughness by a computational/experimental approach for rapid crack propagation in pe pipe. *Int J Fract* 2000;101(3):251–68.
- [9] Ivankovic A, Venizelos G. Rapid crack propagation in plastic pipe: predicting full-scale critical pressure from s4 test results. *Eng Fract Mech* 1998;59(5):607–22.
- [10] Greenshields C, Venizelos G, Ivankovic A. A fluid-structure model for fast brittle fracture in plastic pipes. *J Fluids Struct* 2000;14(2):221–34.
- [11] Williams J, Venizelos G. A perturbation analysis of rapid crack propagation in pressurised pipe. *Int J Fract* 1999;94(2):161–76.
- [12] Beguelin P, Fond C, Kausch HH. Fracture mechanics at intermediate rates of loading: the influence of the acceleration on compact tension tests. *J Phys IV* 1997;7(C3):867–72.
- [13] Beguelin P, Fond C, Kausch HH. The influence of inertial effects on the fracture of rapidly loaded compact tension specimens. Part a: loading and fracture initiation. *Int J Fract* 1998;89(1):85–102.
- [14] Ferrer JB, Fond C, Arakawa K, Takahashi K, Beguelin P, Kausch HH. The influence of crack acceleration on the dynamic stress intensity factor during rapid crack propagation. *Lett Fract Micromech* 1998;87(3):77–82.
- [15] Yoffe EH. The moving griffith crack. *Phil Mag* 1951;42(7):739–50.
- [16] Broberg KB. The propagation of a brittle crack. *Arkiv for Fysik* 1960;18:159.
- [17] Freund LB. Crack propagation in an elastic solid subjected to general loading-i. constant rate of extension. *J. Mech. Phys. Solids* 1972;20:129–40.
- [18] Cros PE, Rota L, Cottenot CE, Schirrer R, Fond C. Experimental and numerical analysis of the impact behaviour of polycarbonate and polyurethane multilayer. *J Phys IV* 2000;10(Pr9):671–5.
- [19] Irwin G, Dally J, Kobayashi T, Fournery W, Etheridge M, Rossmanith H. On the determination of $\dot{a}-k$ relationship for birefringent polymers. *Exp Mech* 1979;19:121–8.
- [20] Sheng J, Zhao YP. Two critical crack propagating velocities for pmma fracture surface. *Int J Fract* 2000;98:L9–14.
- [21] Rosakis A, Zehnder A. On the dynamic fracture of structural metals. *Int J Fract* 1985;27:169–86.
- [22] Ponson L, Bonamy D. Crack propagation in brittle heterogeneous solids: material disorder and crack dynamics. *Int J Fract* 2010;162(1–2):21–31.
- [23] Fineberg J, Bouchbinder E. Recent developments in dynamic fracture: some perspectives. *Int J Fract* 2015;196(1–2):33–57.
- [24] Kopp JB, Schmittbuhl J, Noel O, Lin J, Fond C. Fluctuations of the dynamic fracture energy values related to the amount of created fracture surface. *Eng Fract Mech* 2014;126:178–89.
- [25] Kopp JB, Schmittbuhl J, Noel O, Fond C. A self-affine geometrical model of dynamic rt-pmma fractures: implications for fracture energy measurements. *Int J Fract* 2015;193(2):141–52.
- [26] Fond C, Schirrer R. Dynamic fracture surface energy and branching instabilities during rapid crack propagation in rubber toughened pmma. *Notes au CRAS, Series IIb* 2001;329(3):195–200.
- [27] Fond C, Schirrer R. Influence of crack speed on fracture energy in amorphous and rubber toughened amorphous polymers. *Plast, Rubb Compos, Macromol Eng* 1997;30:116–24.
- [28] Fond C, Schirrer R. Fracture surface energy measurement at high crack speed using a strip specimen: application to rubber toughened pmma. *J Phys IV* 1997;7(C3):969–74.
- [29] Sharon E, Fineberg J. Microbranching instability and the dynamic fracture of brittle materials. *Phys Rev B – Condens Matter Mater Phys* 1996;54(10):7128–39.
- [30] Kopp JB, Lin J, Schmittbuhl J, Fond C. Longitudinal dynamic fracture of polymer pipes. *Eur J Environ Civil Eng* 2014;18(10):1097–105.
- [31] Bradley W, Cantwell W, Kausch H. Viscoelastic creep crack growth: a review of fracture mechanical analyses. *Mech Time-Depend Mater* 1997;1(3):241–68.