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Giovanna GUTIÉRREZ, Jorge Alberto MEDINA, Gilles REGNIER, Bruno FAYOLLE - Thermal oxidation of clay-nanoreinforced polypropylene - Polymer Degradation and Stability - Vol. 95, n°9, p.1708-1715 - 2010

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Thermal oxidation of clay-nanoreinforced polypropylene

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A B S T R A C T

The thermo-oxidation process at low temperatures for a montmorillonite-nanoreinforced polypropylene (PP) was studied. Experimental aging kinetic data at 100, 80 and 60 °C have been obtained and compared with a computational simulation in which a kinetic model based on the closed loop approach was used. As a result, it has been found that the montmorillonite role is not limited to a role of inert filler in the polymer matrix but induces a slight catalytic effect leading to induction period reduction. This effect has been well simulated by increasing initial hydroperoxyde concentration. The consequences of kinetic control by oxygen diffusion have also been investigated by using micro ATR-FTIR mapping to assess concentration profiles of the oxidation products across the sample thickness. It has been found that the oxidized layer thickness is close to 17 µm for the pure polypropylene whereas it is around 10 µm for the nanocomposite at 100 °C. These profile variations have been attributed to differences in oxygen diffusion coefficient values. Simulations based on the kinetic model including diffusion-reaction coupling describe these profiles well.

Keywords:

Polypropylene
Nanocomposite
Micro ATR-FTIR
Oxidation profiles
Modeling

1. Introduction

Nanocomposite materials are attracting great interest in the search for better performing materials with improved thermal, mechanical, electrical and barrier properties. Using plastics as a matrix in this technology is advantageous due to their capability to improve such properties by incorporating nanofiller quantities lower than the 20% vol. required by conventional fillers such as talc, calcium carbonate, and glass fibers [1–3]. Higher aspect ratios as provided with the use of layer silicates allow stronger interfacial clay/polymer interactions [4–7], justifying the use of such silicates as fillers.

The aging process induced by photooxidation in polymer–matrix nanocomposites has been studied by several researchers. Through those studies it was found that the incorporation of nanoclays does not change the aging mechanism of the composite, but rather decreases its induction time, meaning that the montmorillonite (MMT-O) essentially accelerates the oxidation process [8–10]. In the case of thermo-oxidation there is only a small number of studies on the aging kinetics of nanofilled PP. Work by Lomakin – concerning the tracking of weight loss for a nanocomposite during the thermo-oxidation process at temperatures higher than 250 °C by using thermal analysis [11] – or by Vyazovkin –

applying a classical kinetic approach [12] – or by Bertini et al. in which the thermal degradation of a montmorillonite-reinforced polypropylene is studied by TGA and DTG techniques [13] provided key insight. However, the results of these studies were obtained far above the usual temperature interval for the use of PP. To our knowledge, no studies on the thermo-oxidation mechanisms and kinetics in solid state at moderate temperatures exist in the literature.

Concerning a general strategy for studying the thermal oxidation of organic composites, an analytical methodology based on a classical oxidation scheme has been presented by Verdu [14]. From the pure PP oxidation kinetic modeling point of view, a model including reaction-diffusion coupling has been used to simulate the thermal oxidation in air at 80–150 °C [15,16]. Our aim in this paper is to apply both of these approaches to studying the thermal oxidation kinetics of polypropylene nanocomposites. In spite of its complexity, kinetic modeling of the oxidation remains of great interest for the prediction of end-product lifetime.

The aim of this work is to study the montmorillonite influence on the oxidation kinetics of polypropylene at moderated temperatures from experimental and theoretical points of view. The first step to accomplish this objective is compounding the material, which is one of the most critical stages, since good clay dispersion has to be reached in order to separate the clay sheets. To better split the nanosheets, the clay and coupling agent are pre-mixed. Next, the concentrate is blended with pure polypropylene to obtain the nanocomposite. After a complete material characterization using

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a flexible approach [17], the results from thermo-oxidation experiments are obtained. The first goal is to find possible differences between nanoreinforced and pure polypropylene. The second objective is to apply the kinetic model proposed elsewhere [15] to simulate nanocomposite thermal oxidation and to explore the influence of the oxygen diffusion coefficient changes on oxidation profiles of the sample when oxidation is diffusion controlled. Such influence will also be experimentally studied using a new FTIR imaging tool (see in [Experimental part](#)).

To follow the evolution of the oxidation process over time, infrared micro-spectrometry measurements in transmittance mode are often performed [18,19]. However, its spatial resolution of about 25–30 μm does not permit the measurement of oxidation layers, whose individual thickness is of the order or even lower than this resolution. To enhance lateral resolution, some experimental alternatives are available, such as the use of synchrotron radiation as a source for FTIR measurements [20]. By reducing the resolution with the use of high refractive index ATR crystals, attenuated total reflectance infrared spectrometry (ATR-FTIR) permitted us to follow the extent of oxidation in aged carbon-filled rubbers and polyethylene samples [21,22]. Several recent publications have used this approach [23–25]. Here, we applied this micro-ATR-FTIR approach to assess the oxidation profiles and the thickness of the oxidized layer by using a Perkin–Elmer SpotLight 300 FTIR Microscope with Spectrum 100 in ATR mode imaging showing spatial resolution of 6 μm .

2. Experimental

2.1. Materials

The matrix used was the Propilco 01R25 propylene–ethylene copolymer supplied by Propilco (MFI 0.8 g/10min). The nanoclay was Nanofil® SE 3000, montmorillonite modified with di-tallow dimethyl quaternary ammonium salt (Specific weight: 1.2 g/cm³) supplied by Süd Chemie AG. The coupling agent was the Polybond® 3200, 1%w maleic anhydride grafted propylene (MFI 115 g/10 min) provided by Cromtom. The proportion of MMT-O and coupling agent was 9.5% weight and 28.5% weight, respectively.

2.2. Composition of the clay reinforced polypropylene

The mixing of the materials was done in an internal Brabender Plasti-Corder PLE 331 mixer at 200 °C and 90 rpm, during 20 min. To achieve optimum dispersion, clay and coupling agent were incorporated, allowed to mix for 2 min, and then the polymer was added until the end of mixing time. The mixing processing conditions were optimized according to an experimental design. The screw speed varied in order to maximize the exfoliation index which was determined from dynamic shear tests in the linear viscoelastic domain [26]. Then, films were pressed and molded at 160 °C. Pure PP and nanocomposite films were obtained with the same processing conditions to ensure a similar pre-oxidation state.

2.3. XRD

WAXS measurements were performed in a Philips X'pert MRD diffractometer employing punctual CuK α radiation at 40 mA and 40 kV within the scanning angle of 2 θ from 1.5 to 20°.

2.4. STEM

Cross sections of films were thinned down below 100 nm with a JEOL EM9100IS ion slicer. Scanning Transmission Electron Microscopy, STEM, images of nanocomposite were obtained by

using a JEOL 7400F SEM equipped with a STEM accessory at 30 kV. Several areas were scanned to ensure that analysis was representative of the sample.

2.5. Oxygen permeability measurement

At standard temperature and humidity, (23 °C, 50% humidity), oxygen permeability was measured in an OX-TRAN Model 2/21 MOCON permeameter, having 5.067 cm² of active area.

2.6. Thermal aging

In order to study only the thermal oxidation of non-stabilized samples in the first approach, stabilizers were extracted using a published method [27]. Extracted samples were exposed in air-circulating ovens at atmospheric pressure. The thickness of the film samples used to study the aging kinetics at 60, 80 and 100 °C was 75 μm . To measure the oxidation profiles, 150 μm -thick films were used.

2.6.1. FTIR analysis

In order to study the growth of carbonyl (CO) bands during thermal oxidation, FTIR measurements were performed by using a Bruker IFS 128 spectrometer in transmission mode, with a resolution of 4 cm⁻¹ between 400 and 4000 cm⁻¹. Concentration of the carbonyl group was determined from the peak absorbance at 1713 cm⁻¹, assuming this peak was assigned to carboxylic acids forming [28,29]. According to Beer–Lambert law, the carbonyl concentration is given by.

$$[\text{CO}]_{av} = \frac{DO_{co}}{\epsilon_{1713} t \rho_{pp}} \quad (1)$$

where $[\text{CO}]_{av}$ is the average carbonyl concentration expressed in mol kg⁻¹, DO_{co} is the 1713 cm⁻¹ peak absorbance, t is the sample thickness (in cm), ρ_{pp} is the polypropylene density (0.90 kg l⁻¹), and $\epsilon_{1713} = 300 \text{ l mol}^{-1} \text{ cm}^{-1}$ is molar absorptivity. The highest studied average carbonyl concentration was fixed to $[\text{CO}]_{av} = 0.3 \text{ mol kg}^{-1}$ because the sample is highly brittle beyond this value [30].

2.7. Micro ATR-FTIR

The oxidation profiles were measured with a Perkin–Elmer Spectrum SpotLight 300 ATR-FTIR microscope. The high refractive index of the germanium crystal and an effective sample-crystal contact together permit a 6 μm spatial resolution [31]. To acquire an image, both the crystal and the attached sample are moved in the plane of the sample while the infrared beam scans different parts of the sample. Spectra were recorded in reflection mode with steps of 1.56 μm , which corresponds to the size of the image sample grid. Micro ATR-FTIR spectra were collected using two scans per pixel in a wavenumber range between 4000 and 750 cm⁻¹. Samples were coated with MECAPREX MA2 epoxy resin to improve sample handling and prevent the brittle oxidized layer from being torn off, and then polished under water with polishing paper.

Multiple micro ATR-FTIR spectra were made in the transverse direction of the resin-PP-resin samples. To obtain the oxidation profiles, the integration of the absorption values at 1713 cm⁻¹ was done along the sample thickness. The measured profiles were corrected using a baseline between 1600 and 1850 cm⁻¹, then treated to convert their values from absorbance to concentration units. The relative optic densities were calculated as the ratio between the corrected optic density values on the sample thickness and the corrected absorbance value on the sample surface.

3. Results

3.1. Material morphology

To determine the morphology of the nanocomposite WAXS analyses were performed. To assess if the proportion of ethylene in the composite was relevant, presence of the beta phase peaks corresponding to ethylene crystalline structures was verified. Only the characteristic alpha phase peaks corresponding to crystalline PP were present at 14.04° (Miller index 110) and 16.9° (Miller index 040). There was no peak at 16° corresponding to the beta phase, meaning that the quantity of beta phase present in the composite was under the sensitivity limit. A volumic crystallinity close to 0.5 was measured for both materials.

A second 2θ angle scanning was performed in order to characterize the morphological state in the clay nanocomposite material by measuring the interlayer spacing of the clay before and after the material processing. The spacing variation between clay layers could be directly correlated to filler dispersion through the polymeric matrix. As shown in Fig. 1, the measured nanoclay basal distance was $d_0 = 2.7$ nm and the distance measured after processing was $d = 3.1$ nm. The difference found between initial and final interlayer distances suggests an intercalated/exfoliated morphology [4]. To qualitatively confirm this affirmation, a STEM study was performed. Fig. 2 shows a STEM image at $120,000\times$ magnification: the dispersion degree of the clay was characteristic of intercalation with some signs of exfoliation. Aggregates composed of 8 clay sheets, in average, are clearly distinguishable at $500,000\times$ magnification (Fig. 3).

Based on both WAXS and STEM results, we conclude that the nanocomposite has an intercalated/exfoliated morphology, which corresponds to the most common mixture state reached in materials prepared by the melt intercalation method [32]. Fully exfoliated morphology could be obtained when the nanocomposite was obtained by *in situ* intercalative polymerization or by intercalation of polymer or pre-polymer from solution methods [17,33].

3.2. Oxygen permeability

We were able to measure the oxygen permeability twice for the pure polypropylene and three times for the nanocomposite, only at room temperature. The mean values were $2.28 \times 10^{-17} \text{ m}^2 \text{ s}^{-1} \text{ Pa}^{-1}$ and $1.17 \times 10^{-17} \text{ m}^2 \text{ s}^{-1} \text{ Pa}^{-1}$ respectively. We observed a small, although statistically insignificant, difference, due to the low

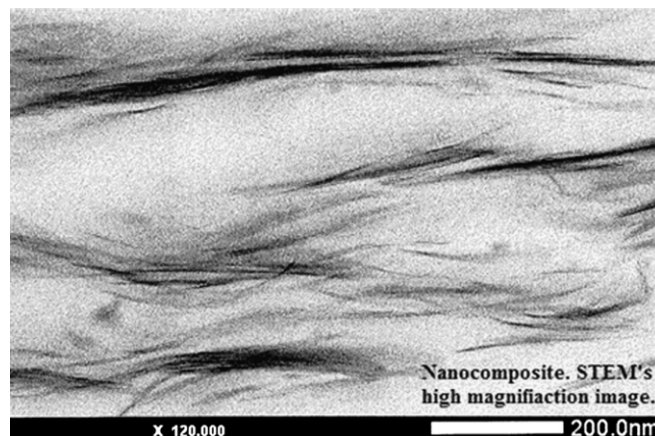


Fig. 2. High magnification STEM image of a film cross section: nanoclay dispersion. 120,000 X photograph.

number of tests. Similar results were recently published for a clay-nanoreinforced polypropylene system which had oxygen permeability improvements between 10 and 40% [34]. Then, due to the small difference between the oxygen permeabilities of pure PP and nanoreinforced PP, the measurement dispersion and the unknown influence of temperature, we applied an inverse method to determine the oxygen diffusion coefficient at 100°C . This issue is widely discussed in Section 4.3.

3.3. Characterization of the oxidation process

IR spectra in the carbonyl region of the matrix (Fig. 4a) and nanocomposite (Fig. 4b) at various oxidation times at 100°C reveal the growth of oxidation products. One notices that an extra peak at 1740 cm^{-1} is initially present in the nanocomposite spectra.

It can be attributed to the presence of saturated aliphatic esters and γ -lactones ($1735\text{--}1750 \text{ cm}^{-1}$) due to the organic modification of the montmorillonite.

The thermo-oxidation kinetics were studied at 60, 80 and 100°C by following the carbonyl appearance as a function of time for the polypropylene and the nanocomposite (Fig. 5). We note that the resulting curves have the typical shape for a given temperature regardless of polymer identity; that is, all curves display an induction stage followed by a short period during which oxidation

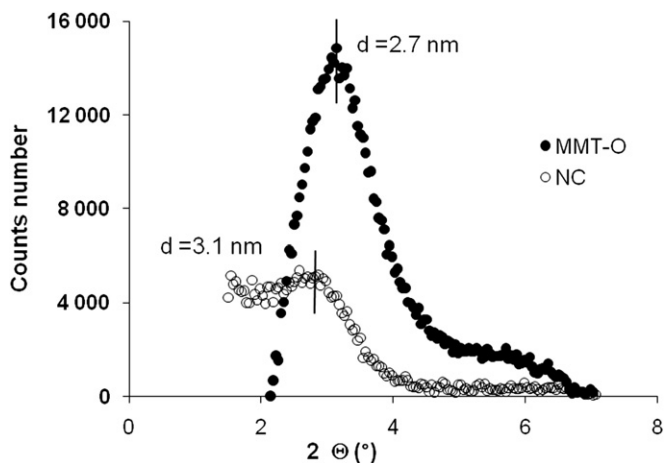


Fig. 1. Differences of count numbers in WAXS tests between pure polypropylene and composite samples.

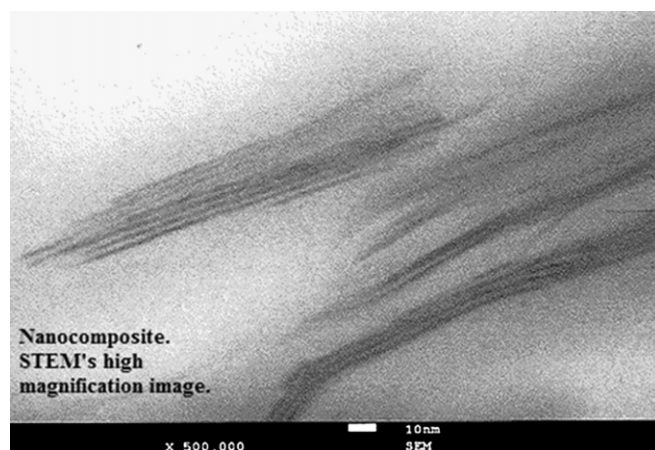


Fig. 3. High magnification STEM image of a film cross section: nanoclay dispersion. 500,000 X photograph.

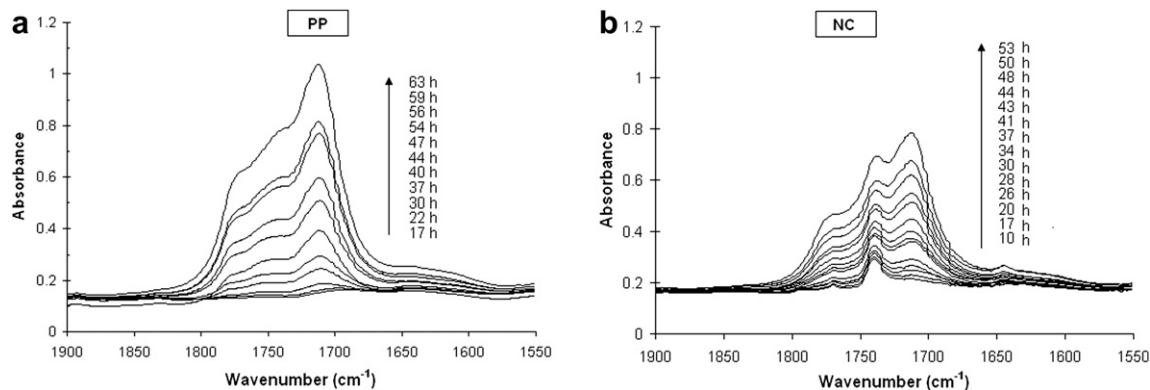


Fig. 4. Evolution of the carbonyl group concentration at 1713 cm^{-1} and temperature fixed at $100\text{ }^{\circ}\text{C}$. a) polypropylene, b) nanocomposite.

is strongly auto-accelerated and a steady-state period in which the oxidation rate is almost constant. The induction time (t_i) is commonly used to characterize the oxidation rate. The t_i value was experimentally determined by the point at which the straight line corresponding to the steady-state crossed the t -axis.

Induction time values for both polymers at the three defined temperatures are included in Table 1. It appears that the induction time values obtained for the nanocomposite are lower for all exposure temperatures, indicating that the nanocomposite is slightly more easily oxidized. The potential causes of this behavior are discussed below.

In the first approach, it was necessary to assess the significance of the induction time difference between both polymers. For this purpose, we plotted in an Arrhenius diagram the induction times of polypropylene and the nanocomposite with compiled literature data (Fig. 6) [35–51]. From this coarse grain point of view, it clearly appears that this difference observed in Table 1 between pure and nanocomposites is not very significant since all induction period values are within the scatter of points corresponding to induction period values reported in the literature. As a result, the nanocomposite can be considered as a pure polypropylene for its oxidation behavior according to this approach. Lastly, we concluded also that the nanofiller has no influence on the activation energy, its average value being close to 116 kJ mol^{-1} .

However, the fact that smaller induction times were systematically found for the nanoreinforced materials led us to investigate possible causes to explain the difference between both polymers under study. To explore this divergence more precisely, the carbonyl concentration was plotted as a function of time in a linear scale for an exposure temperature of $100\text{ }^{\circ}\text{C}$ (Fig. 7). It is noteworthy that if the nanocomposite exhibits a lower induction period as we

have seen before, the nanocomposite also exhibits a lower oxidation rate in the steady-state.

3.4. Experimental measurement of the oxidation profiles

Fig. 8a and b show carbonyl profiles for the pure and nanocomposite polypropylene, respectively. On the y axis were plotted the relative absorbances that are the ratio between the absorbance on each point at 1713 cm^{-1} and the absorbance on the surface of the sample at 1713 cm^{-1} . Such values were corrected by multiplying each absorbance value by the ratio between the absorbance value on the sample surface for $[\text{CO}]_{av} = 0.3\text{ mol kg}^{-1}$ obtained from the simulation and the experimental absorbance value on the sample surface at $[\text{CO}]_{av} = 0.3\text{ mol kg}^{-1}$. The simulation values were previously corrected by adding the carbonyl concentration measured in transmission mode at time zero before starting the aging process.

It clearly appears that for both materials (Fig. 8) the thickness of the oxidized layer is close to $15\text{ }\mu\text{m}$, which means that the oxidation is heterogeneous after the end of the induction period and that the oxidation kinetics are oxygen diffusion limited. To our knowledge, it is the first time that an oxidation profile for PP was measured with such accuracy.

By comparing Fig. 8a and b, it appeared that for the same $[\text{CO}]_{av} = 0.3\text{ mol kg}^{-1}$ for instance, the nanocomposite showed a thickness of oxidized layer (TOL) close to $10\text{ }\mu\text{m}$ whereas the pure PP TOL is around $17\text{ }\mu\text{m}$. Such a difference could be related to the oxygen diffusion. To explore the influence of the oxygen diffusion coefficient on the shape of the oxidation profiles, and especially on the thickness of the oxidation layer, we used an analytical approach which is discussed in Section 4.

4. Discussion

To model oxidation behavior of the clay reinforced material, it is necessary to consider the montmorillonite dual influence on the oxidation phenomena. First, the montmorillonite can modify the oxidation mechanism. Second, montmorillonite can affect material physical properties, in particular, the oxygen diffusion coefficient

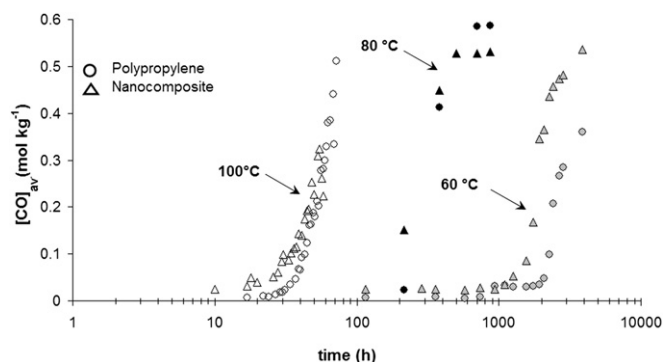


Fig. 5. Kinetic curves of carbonyl build up at 60 , 80 and $100\text{ }^{\circ}\text{C}$ for a) pure polypropylene, and b) nanocomposite.

Table 1
Oxidation induction times for the polypropylene and composite films.

$T\text{ (}^{\circ}\text{C)}$	$t_i\text{PP (h)}$	$t_i\text{NC (h)}$
100	33	25
80	220	150
60	1580	1300

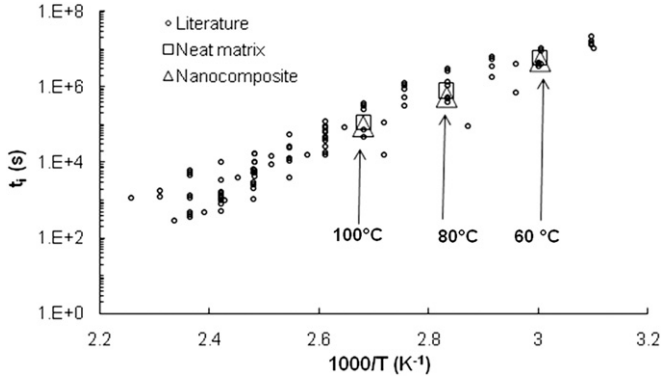


Fig. 6. Arrhenius dependence for induction times (t_i) at 60, 80 and 100 °C for the pure polypropylene and nanocomposite.

and the reaction-diffusion coupling. Both chemical and physical influences of montmorillonite are studied in this work. Section 4.1 presents the kinetic model used to simulate the oxidation behavior of the nanocomposite. In Section 4.2, model parameters and comparison between the model predictions and experimental results are discussed. Section 4.3 is dedicated to the discussion of the influence of oxygen diffusion coefficient variations on oxidation profiles as a function of sample thickness.

4.1. Kinetic modelling

The Closed Loop Model (CLM) was used to model oxidation kinetics [16]. The model is derived from a standard mechanistic scheme in which oxidation results from a chain reaction initiated by hydroperoxide decomposition. Alkyl (P^\bullet) and peroxy (PO_2^\bullet) radicals are the chain carriers. The three possible terminations between these radicals are taken into account. The predictive power of this model is now well established [14–16].

For the general case where termination may occur by bimolecular reactions, the closed-loop mechanistic scheme involves seven elementary reactions:

Initiation

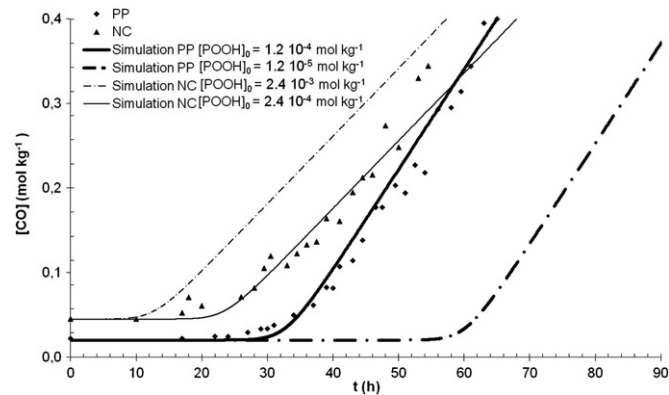
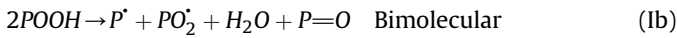
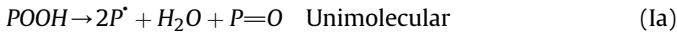
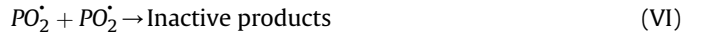


Fig. 7. Comparison between experimental data and simulation kinetic curves at 100 °C and $[POOH]_0$ influence on kinetic curves: $[POOH]_{0NC}$: 2.4×10^{-3} and 2.4×10^{-4} mol kg^{-1} , $[POOH]_{0PP}$: 1.2×10^{-4} and 1.2×10^{-5} mol kg^{-1} .

Propagation



Termination



A set of ordinary differential equations is derived from this scheme; it expresses the concentration changes of the involved reactive species. Equations (2)–(4) correspond to the alkyl radical (P^\bullet), peroxy radical (PO_2^\bullet), and hydroperoxide ($POOH$) concentrations respectively, and k_{1u} , k_{1b} , k_2 , k_3 , k_4 , k_5 , and k_6 are rate constants for reactions (I)–(VI).

$$\frac{d[P^\bullet]}{dt} = 2k_{1u}[POOH] + k_{1b}[POOH]^2 - k_2[P^\bullet][O_2] + k_3[PO_2^\bullet][PH] - 2k_4[P^\bullet]^2 - k_5[P^\bullet][PO_2^\bullet] \quad (2)$$

$$\frac{d[PO_2^\bullet]}{dt} = k_{1b}[POOH]^2 + k_2[P^\bullet][O_2] - k_3[PO_2^\bullet][PH] - k_5[P^\bullet][PO_2^\bullet] - 2k_6[PO_2^\bullet]^2 \quad (3)$$

$$\frac{d[POOH]}{dt} = -k_{1u}[POOH] - 2k_{1b}[POOH]^2 + k_3[PO_2^\bullet][PH] \quad (4)$$

The initial and boundary conditions input to solve the system of differential equation were $[PO_2^\bullet] = 0$, $[P^\bullet] = 0$, $[POOH] = [POOH]_0$, $[PH] = [PH]_0$, $[O_2] = [O_2]_s$, where $[PH]_0$ and $[O_2]_s$ are the initial concentrations of the $[PH]$ and $[O_2]$ species respectively (see Table 3).

The oxygen diffusion effect could also be included in the CLM by adding the oxygen diffusion term (Equation (5)). D_{O_2} is the coefficient of oxygen diffusion in the polymer. This coefficient is lower in the nanocomposite than in the pure matrix.

$$\frac{d[O_2]}{dt} = D_{O_2} \frac{\partial^2 [O_2]}{\partial x^2} - k_2[P^\bullet][O_2] + k_6[PO_2^\bullet]^2 \quad (5)$$

The CLM input is completed by adding an extra condition related to the oxygen diffusion: at any time, $[O_2] = [O_2]_s$ for $x = -L/2$ and $x = L/2$, the origin of depth coordinates being taken in the middle of the sample thickness.

Assuming that the source of the carbonyl group (CO) is the decomposition of hydroperoxides, carbonyl concentration $[CO]_{av}$ is written as:

$$\frac{d[CO]_{av}}{dt} = [k_{1u}[POOH] + k_{1b}[POOH]^2] \times [1 - X_c - MMT] \times \rho_{PP}^{-1} \quad (6)$$

where X_c is the crystallinity ratio of the sample expressed in volume fraction ($X_c = 0.5$), and MMT is the volume fraction of montmorillonite equal to 0.08 in our case. Since oxidation occurs only in the amorphous phase, the X_c and MMT terms were included in Equation (6) as correction factors related to the amorphous phase fraction present in the polymer in which oxidation occurs. The

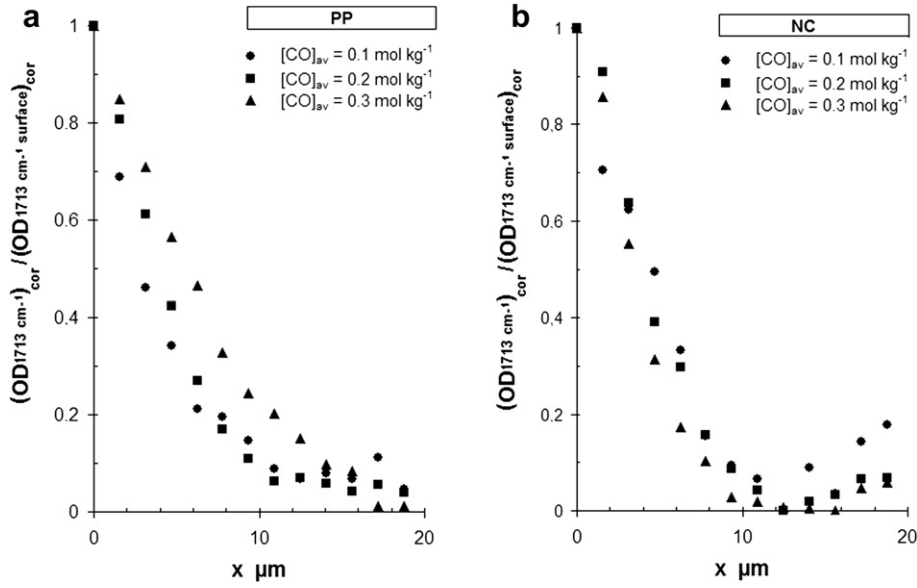


Fig. 8. Relative optic density (OD) at 1713 cm^{-1} of the experimental oxidation profiles at $[\text{CO}]_{av} = 0.1, 0.2$ and 0.3 mol kg^{-1} for a) pure polypropylene b) nanocomposite.

kinetic parameters' values for pure polypropylene at $100\text{ }^{\circ}\text{C}$ reported by Richaud [52] are shown in Table 2.

4.2. Carbonyl build-up simulation

The initial hydroperoxide concentration $[\text{POOH}]_0$ can be considered as the initial POOH concentration kinetically equivalent to the concentration of all the radical sources initially present in the material. Its value is generally lower than the sensitivity threshold of available titration methods. It appears thus as an adjustable parameter of the kinetic model. To illustrate the sensitivity of the model to changes of this parameter, simulated carbonyl growth curves are included in Fig. 7 adjusting $[\text{POOH}]_0$ values for both pure and reinforced polypropylene to fit experimental carbonyl growth curves in Fig. 7. After optimization, we selected for the pure and reinforced polypropylene $[\text{POOH}]_{0PP} = 1.2 \times 10^{-4}\text{ mol kg}^{-1}$, and $[\text{POOH}]_{0NC} = 2.4 \times 10^{-4}\text{ mol kg}^{-1}$ respectively to achieve a good agreement between the experimental and simulated CO values (Table 3).

These results can be interpreted as follows: Montmorillonite exerts a direct (catalytic effect by metallic impurities) or indirect (thermomechanic history during processing) small influence on nanocomposite pre-oxidation, leading to the reduction of the induction time. A similar effect has been observed from the presence of residual Ziegler–Natta catalyst Titanium [53]. As a result, the initial hydroperoxide concentration $[\text{POOH}]_0$ is slightly higher in the nanocomposite than in the pure polymer, which explains the observed difference in the oxidation behavior. It is noteworthy that a slight catalytic effect of montmorillonite on PP photooxidation was already reported in the case of intercalated and exfoliated samples [10,54].

The steady-state behavior is well simulated without parameter modifications (Fig. 7). The disparity between PP and the nanocomposite is only due to the difference in the amorphous phase

concentration (see Equation (6)). We recall that in a closed-loop process, the steady-state behavior is independent of the initiation rate and the $[\text{POOH}]_0$ value. Experimental observations are thus in good agreement with theoretical prediction.

4.3. Oxidation controlled by diffusion

When the oxidation process is controlled by oxygen diffusion, oxidation rate and oxygen diffusion competition leads to an oxidation profile formation. In other words, oxidation is likely confined to a superficial layer when film thickness is higher than $20\text{ }\mu\text{m}$ as predicted by [15]. According to the measured oxidation profiles (Fig. 8a and b), there is a difference between the thickness of the pure polypropylene and the oxidized nanocomposite layer. In that case, it should be possible to describe the oxidation behavior of both materials using the oxygen diffusion coefficient as fitting variable, which will be different for each material. Since permeation tests displayed relatively high uncertainty, we decided to determine the oxygen diffusion coefficients of pure and nanoreinforced polypropylene from the kinetic model, using an inverse method.

Fig. 9 illustrates the influence of oxygen diffusion coefficient value on the oxidation profile shape at $100\text{ }^{\circ}\text{C}$ and a $[\text{CO}]_{av}$ equal to 0.3 mol kg^{-1} for the pure polypropylene. Several oxygen diffusion coefficient values D_{O_2} ranging from $2 \times 10^{-13}\text{ m}^2\text{ s}^{-1}$ and $2.0 \times 10^{-11}\text{ m}^2\text{ s}^{-1}$ were used in simulation. It is noticeable that for such D_{O_2} values, the profile shape radically changes: high D_{O_2} generates wide profiles, but for smaller values of D_{O_2} the obtained oxidation profiles are narrowed. It is clear that the kinetic model is sensitive enough to discern small changes in D_{O_2} and their influence on the shape of the oxidation profiles. In this way it was possible to identify the oxygen coefficient value for each of the materials, and $D_{O_2} = 2.0 \times 10^{-12}\text{ m}^2\text{ s}^{-1}$ and $1.2 \times 10^{-12}\text{ m}^2\text{ s}^{-1}$ were found for the polypropylene and the nanocomposite respectively (Table 3).

Table 2
Kinetic constants of the kinetic closed loop model.

$T\text{ (}^{\circ}\text{C)}$	$k_{1u}\text{ (s}^{-1}\text{)}$	$k_{1b}\text{ (1 mol s}^{-1}\text{)}$	$k_2\text{ (1 mol s}^{-1}\text{)}$	$k_3\text{ (1 mol s}^{-1}\text{)}$	$k_4\text{ (1 mol s}^{-1}\text{)}$	$k_5\text{ (1 mol s}^{-1}\text{)}$	$k_6\text{ (1 mol s}^{-1}\text{)}$
100	1.5×10^{-7}	1.0×10^{-4}	1.3×10^7	7.0×10^{-1}	1.0×10^{10}	2.6×10^9	3.0×10^3

Table 3
Material parameters for the kinetic closed loop model.

	$[PH]_0$ (mol l ⁻¹)	$[POOH]_0$ (mol l ⁻¹)	X_c	MMT-O	S_{O_2} (mol l ⁻¹ Pa ⁻¹)	D_{O_2} at 100 °C (m ² s ⁻¹)
PP	20	1.2×10^{-4}	0.5	0	1.15×10^{-7}	2.0×10^{-12}
NC	20	2.4×10^{-4}	0.5	0.1	1.15×10^{-7}	1.2×10^{-12}

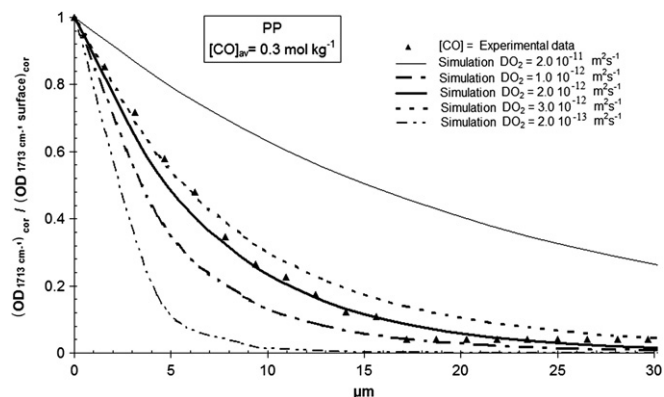


Fig. 9. Influence of the oxygen diffusion coefficient D_{O_2} variation on the simulated carbonyl distribution in the thickness of 150 μm -thick PP films at $[CO]_{av} = 0.3 \text{ mol kg}^{-1}$.

Fig. 10 shows the theoretical and experimental oxidation profiles in mol kg^{-1} for three different exposure times, corresponding to the average concentration in the whole thickness of the film $[CO]_{av} = 0.1, 0.2$, and 0.3 mol kg^{-1} . These concentrations were measured in transmission mode using the Beer–Lambert law. The integration values of $[CO]_{av}$ along the 150 μm -sample thickness were taken from the y-axes of Fig. 10a and b. Experimental values were treated using a single adjusting factor equal to the experimental absorbance in the sample thickness to the simulated $[CO]_{av}$ on the sample surface for an average carbonyl concentration of $[CO]_{av} = 0.3 \text{ mol kg}^{-1}$ Fig. 10a and b display the last 20 μm (from the surface) of pure and nanocomposite polypropylene cross sections, respectively.

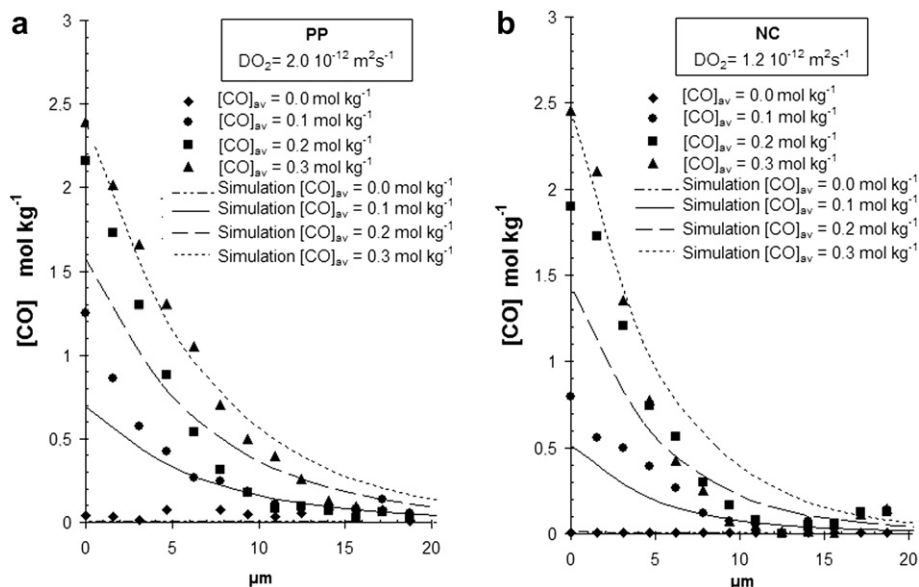


Fig. 10. Theoretical and experimental oxidation profiles in pure and reinforced polypropylene films for $[CO]_{av}$ equal to 0.1, 0.2, 0.3 mol kg^{-1} : a) pure polypropylene, simulated with $D_{O_2,PP} : 2.0 \times 10^{-12} \text{ m}^2 \text{ s}^{-1}$, b) Nanocomposite, simulated with $D_{O_2,NC} : 1.2 \times 10^{-12} \text{ m}^2 \text{ s}^{-1}$.

As shown in Fig. 10, a good agreement between simulation and experimental data was obtained by using for pure polypropylene, $D_{O_2,PP} : 2.0 \times 10^{-12} \text{ m}^2 \text{ s}^{-1}$ and for nanocomposite, $D_{O_2,NC} : 1.2 \times 10^{-12} \text{ m}^2 \text{ s}^{-1}$. Some noise was found in the profile simulation of the nanocomposite; this is attributed to mixing effects related to the clay dispersion in the matrix and to the quality of the sample surface when the ATR testing was performed.

Since the TOL was 17 μm for the PP and 10 μm for the nanocomposite, it is important to point out that the TOL results from the competition between the oxidation rate and O_2 diffusion (see kinetic model where the differential equations included oxidation and Fick O_2 diffusion). However, there is no theoretical reason to assume that nanofillers increase the oxidation rate since this property is linked to the propagation (k_3) and termination (k_6) steps of the oxidation mechanism and the amorphous phase content. This may be why nanofillers act only as “defects,” leading to a decreased induction period (increase of $[POOH]_0$ in the modeling). Furthermore, in Fig. 7, the experimental oxidation rate corresponding to the slope of $[CO]_{av}$ build-up after the induction period measured on thin films (75 μm) shows a decrease in the case of NC compared to PP which is explained by the decrease of the amorphous phase content. At last, from the modeling point of view, an oxidation rate increase (by increasing k_3 for instance) does not contribute to a good description of the oxidation profiles. As a result, the simulated profile curves are unique for a good fit of all experimental data (carbonyl build-up and profiles).

5. Conclusions

Thermal oxidation of pure polypropylene and its nanocomposite was studied experimentally and simulated with a kinetic model. The first objective of this study was to determine possible effects of montmorillonite on the oxidation kinetic behavior of polypropylene at moderate temperatures. Slight differences between induction times of polypropylene and its nanocomposite were found. They can be attributed to a presumed catalytic effect of montmorillonite impurities on PP oxidation or simply to a difference in thermomechanical histories between PP and the nanocomposite. From a modeling point of view, this effect can be taken

into account by increasing initial hydroperoxide concentration $[POOH]_0$. As a result, montmorillonite could be considered as a filler which leads an oxidation rate increase only during the induction period.

The second objective was to study the physical influence of montmorillonite in polypropylene and its effect on oxidation profiles. By using the ATR-FTIR microscope, evidence was obtained that the oxidized layer thickness is close to 17 μm for pure polypropylene and 10 μm for the nanocomposite for an exposure at 100 °C. This difference has been attributed to the decrease of the oxygen diffusion coefficient by adding montmorillonite. This hypothesis was examined by simulating carbonyl profiles. A good agreement between experimental and simulated profiles was obtained by using an oxygen diffusion coefficient value (D_{O_2}) for the nanocomposite 40% smaller than that for pure polypropylene.

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